

# Strain induced Electric Effects in Materials

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In this work, we put forward *Yuheng Zhang equation*  $\nabla E_F(\vec{r}) = qE(\vec{r})$ , a newly found law for condensed materials, where  $E_F(\vec{r})$  is position dependence of Fermi surface energy,  $q$  is electron charge and  $E(\vec{r})$  is the induced electric field. Utilizing this law, we mainly investigate the mechanical-electric coupling effect in materials, which is named as *Yuheng Zhang effect*. Through investigations, it is found that 1) under strain gradient, materials could behave as a p-n junction, exhibiting current-rectifying properties; 2) thermoelectric power or Seebeck coefficient of materials is contributed by thermal expansion (contraction) and *Yuheng Zhang effect*, which may be described by *Yuheng Zhang coefficient*; 3) this effect gives a new mechanism of electron-phonon interaction; 4) an electric field always accompanies defects in materials, such as dislocations, Abrikosov vortices and so on, and their electric field and related electrical potential are given; 5) strain gradient can cause an electric polarization in dielectric materials; 6) the gravity induced electric field persists within planets including the earth; 7) *Yuheng Zhang effect* may be the microscopic physical origin of flexoelectric effect and the flexoelectric coefficients can be derived by means of *Yuheng Zhang equation*; 8) microscopic theory of shock polarization, another long-standing problem in the world, may be clarified in terms of *Yuheng Zhang effect*. In all, *Yuheng Zhang equation* and *Yuheng Zhang effect* offer people new understanding of electric properties of strained materials, and may find various applications in multi-areas.

## Introduction

*Yuheng Zhang effect*, i.e., strain-induced electric effect in metals, has been proposed recently [1]. It gave several interesting predictions and was employed to account for source of the earth's geomagnetic field [2]. As stated in reference [1], *Yuheng Zhang effect* may be applicable in metals, but one may naturally ask a question whether this effect applies for semiconductors and insulators or not.

In this work, we investigate the situations for semiconductors and insulators and try to clarify whether *Yuheng Zhang effect* is valid for these materials or not.

## Discussions

Like the case in metals, let us consider a typical semiconductor or insulator undergoing strains. For the regions under strains, the Fermi surface for electrons may be lifted. Thus a discrepancy of Fermi surface appears between the strained and unstrained regions. The electrons in regions with higher Fermi surface tend to shift to regions with lower Fermi surface, thereby creating an electric field and an electric potential between the regions, as is shown in Figure 1. As is known, mass of electrons is much smaller than that of ions or atoms in materials, so response time scale of electrons is much shorter than that of ions or atoms. Therefore once the ions or atoms move in the materials, electrons response rapidly and almost approach their new equilibrium state at instant. These processes in semiconductors and insulators may be described by the equation as in metals [1],

$$\nabla E_F(\vec{r}) = q\vec{E}(\vec{r}) \quad (1)$$

$$qV_Z = E_F(\vec{r}_1) - E_F(\vec{r}_2) \quad (2)$$

where  $E_F(\vec{r})$  is position dependence of Fermi surface energy,  $q$  is electron charge and  $\vec{E}(\vec{r})$  is the yielded electric fields,  $V_Z$  is the induced electric potential. Despite that this equation first appears in reference [1], to be consistent with name of this effect, here Equation (1) is formally named as *Yuheng Zhang Equation* and the potential described by Equation (2) could be called *Yuheng Zhang potential*.

*Yuheng Zhang Equation* may be valid for metals, semiconductors and insulators in solid and liquid state (In liquid state, Fermi surface energy may be chemical potential.), regardless of their dimensions, one, two or three.

Let us examine the similarity and difference between *Yuheng Zhang effect* and piezoelectric effect. Their similarity is that both of them happen due to strains caused by applied stress and they can generate separation of accumulated charges in some semiconductors and insulators [3]. However, their difference is obvious. First, rigorously speaking, *Yuheng Zhang effect* is induced by strain gradient in materials rather than uniform strains for piezoelectric effect; second, *Yuheng Zhang effect* might be valid for all condensed materials including crystalline materials with center-inversion symmetry, but piezoelectric effect only exist in insulating materials without center-inversion symmetry; third, *Yuheng Zhang effect* may applies for metals, however, piezoelectric effect does not.

*Yuheng Zhang potential* is also compared with Galvani potential here. In some respect, *Yuheng Zhang potential* is like Galvani potential. Galvani potential occurs when two different materials contact each other, and it is given by difference between Fermi surface energies of the two materials [4],  $qV = E_{F2} - E_{F1}$ , where  $q$  is electron

charge,  $V$  is the Galvani potential,  $E_{F1}$  and  $E_{F2}$  are the Fermi surface energy of the two different materials, respectively. If the material under different strains are regarded as different materials, *Yuheng Zhang potential* is the same as Galvani potential. But they differ from each other. *Yuheng Zhang potential* arises from strain-induced lift of Fermi surface energy of the identical material, whereas Galvani potential comes from the heterogeneous junction of two distinct materials with different Fermi surface energies.

Generally speaking, many methods could alter Fermi surface energy. As usually encountered in semiconductors, doping can change Fermi surface energy, e.g., donor doping and acceptor doping in silicon. Besides, temperature variations and magnetic field also enable Fermi surface energy lift, despite very small alteration in many materials. The related physical processes are interesting and might be described by *Yuheng Zhang equation* and *Yuheng Zhang potential*, but in this work we only focus on strain effect.

Like in metals [1], this effect in semiconductors and insulators may also lead to several interesting phenomena.

### **(1) P-n junction behaviors**

For materials under strain gradient, especially metal and narrow band semiconductors, may behave as an ideal p-n junction, displaying current-rectifying behaviors because of *Yuheng Zhang potential* as shown in Figure 2. And the biased electrical voltage  $V_Z$  equals to *Yuheng Zhang potential*. This property may be useful in some situations. For example, one must be careful when performing resistance measurement of materials experiencing non-uniform strain. As shown in Figure 2(b)

(black line), the measured resistance might be larger than the actual values.

## (2) Thermoelectric power

*Yuheng Zhang coefficient* [1] may be also valid for semiconductors and insulators,

$$S_Z = \frac{\alpha_V V(T)}{q} \frac{\delta E_F(\bar{r})}{\delta V(T)} \quad (3)$$

where  $q$  is electron charge,  $V(T)$  is temperature  $T$  dependence of volume and  $\alpha_V$  is volume expansion parameter. In measurements on Seebeck coefficients, it was previously believed that Seebeck coefficient was summation of phonon drag component  $S_p$  and hot electron diffusion thermopower  $S_d$ , i.e.,  $S=S_p+S_d$ . Here we point out that experimental result of Seebeck coefficient may be a summation of *Yuheng Zhang coefficient*  $S_Z$ , phonon drag component  $S_p$  and hot electron diffusion thermopower  $S_d$ . It gives

$$S = S_Z + S_p + S_d \quad (4)$$

Sometimes *Yuheng Zhang coefficient* may dominate temperature dependence of Seebeck coefficient, and its sign relies on both the thermal expansion parameter and strain dependence of Fermi surface energy. This may give a new understanding for temperature dependence of thermoelectric power of materials.

## (3) New source of electron-phonon interaction

Strains, including elastic and plastic strains, usually come from motion and redistribution of ions in materials. When ions move the electrons can quickly follow them and reach a new equilibrium state due to much smaller electron mass than ions. Consider a case that a longitudinal wave passing through a simple cubic crystal, due to induced periodic displacement of ions, a periodic strain gradient may appear and

therefore *Yuheng Zhang effect* occurs, generating an electric field and a related electrical potential, as is shown in Figure 3 (b). Electrons will experience the electric field and their potential energy is correspondingly lifted. This is the newly proposed mechanism of electron-phonon ( $e$ - $p$ ) interaction in this work. By means of similar calculations [5], its mathematic expression may be written as

$$H_{e-p} = -i \sum_{\vec{k}, \vec{q}, \lambda} \vec{\lambda}_q \cdot \vec{q} \sqrt{\frac{\hbar}{2MN\omega_{q,\lambda}}} E_F(\vec{q}) (b_{\vec{q},\lambda} + b_{-\vec{q},\lambda}^+) C_{\vec{k}}^+ C_{\vec{k}-\vec{q}} \quad (5)$$

where  $H_{e-p}$  is  $e$ - $p$  Hamiltonian,  $b_{q,\lambda}$  and  $b_{q,\lambda}^+$  are annihilation and creation operators for  $\lambda$ -branch phonons with momentum  $q$ ,  $C_{\vec{k}}$  and  $C_{\vec{k}}^+$  are annihilation and creation operators for electrons with momentum  $k$ , respectively,  $\omega_{q,\lambda}$  is phonon energy,  $M$  is ion mass,  $N$  is number of ions in the crystal,  $\vec{\lambda}_q$  is a direction unit vector for ion vibrations, Fermi surface energy for electrons at ion position  $\vec{R}_j$  is  $E_F(\vec{r} - \vec{R}_j)$ , and its Fourier form is  $E_F(\vec{r} - \vec{R}_j) = \frac{1}{N} \sum_{\vec{k}} E_F(\vec{k}) e^{i\vec{k} \cdot (\vec{r} - \vec{R}_j)}$ .

This new  $e$ - $p$  interaction is very like the conventional  $e$ - $p$  interaction [5] in formalism, but they are distinct in physics. This  $e$ - $p$  interaction comes from *Yuheng Zhang effect*, but the conventional  $e$ - $p$  interaction stems from screened coulomb interaction between ions and electrons. Strength of the new  $e$ - $p$  interaction sensitively depends on functional derivative of  $\delta E_F / \delta \mathcal{C}$  ( $\mathcal{C}$  is position dependence of stain, which is negative for compressed strain and positive for tensile strain) and can be more easily determined by first-principle calculations. Like the conventional  $e$ - $p$  interaction, the new  $e$ - $p$  interaction may play a vital role in our understanding multi-physical properties of crystals.

At the end of this part, let us give a predicted interesting phenomenon. Suppose a crystal is periodically strained along a direction and we measure its resistance along the direction, as is shown in Figure 3(b). According to previous discussions, the measured electrical current may be

$$I_m = \frac{1}{R_0} \left( U_e - \frac{LV_Z}{L_0} \right) \quad (6)$$

where  $I_m$  is measured electrical current flowing through the crystal,  $R_0$  is actual resistance,  $U_e$  is applied external electrical voltage,  $L$  is length of the crystal,  $L_0$  is periodic strain length,  $V_Z$  is the induced *Yuheng Zhang potential* in a period as shown in Figure 3(c). It can be seen that the measured resistance is always larger than its actual value. A shorter period and larger strain will cause a more notable increase of measured resistance. On the contrary, the measured resistance decreases if the external voltage increases, which is very like electrical behaviors of charge density wave [6].

#### **(4) Defects in materials**

##### **i) Dislocations**

During plastic deformation of materials such as sodium chloride, electric charges appear and they are reported to accompany with dislocations [7, 8]. For a small angle grain boundary constructed by dislocations, experimental investigations show that it can bear charges and migrate under an external electric field [9]. Further experimental observations indicate that lattice-mismatch induced dislocations in a thin film give rise to a drastic degradation in piezoelectric constant and switched electric polarization [10]. Phenomenological analysis demonstrated that in ferroelectric films polarization varies dramatically near a dislocation and its gradient leads to suppression of polarization over

several nanometers [11]. However, the microscopic mechanism has been still unclear [12].

In this part, the involved phenomena and their microscopic physical origin is explored in terms of *Yuheng Zhang effect*.

Let us consider edge and screw dislocations in an isotropic material. According to elastic stress field of an edge dislocation [13] and Hooke's law, position dependence of local volume under the stress field of the dislocation is  $V(x, y) = V_0 \left[ 1 - \frac{(1-2\nu)b_e}{2\pi(1-\nu)} \frac{y}{x^2+y^2} \right]$ , where  $\nu$  is Poisson's ratio,  $V(x, y)$  is position  $(x, y)$  dependence of volume,  $V_0$  is unstrained volume,  $b_e$  is Burgers vector of an edge dislocation in  $x$  direction. Ignoring boundary effect, e.g., image dislocation induced strain relaxation, based on Equation (1) and (2), the caused electric field around a single edge dislocation and electrical potential between any two points are easily obtained

$$\vec{E}(x, y) \approx \frac{V_0}{q} \frac{\delta E_F}{\delta V} \frac{(1-2\nu)b_e}{2\pi(1-\nu)} \left[ \frac{2xy}{(x^2+y^2)^2} \vec{x} + \frac{y^2-x^2}{(x^2+y^2)^2} \vec{y} \right] \quad (7)$$

$$V_Z(x_2, y_2) - V_Z(x_1, y_1) \approx \frac{V_0}{q} \frac{\delta E_F}{\delta V} \frac{(1-2\nu)b_e}{2\pi(1-\nu)} \left( \frac{y_2}{x_2^2+y_2^2} - \frac{y_1}{x_1^2+y_1^2} \right) \quad (8)$$

where  $q$  is electron charge,  $\vec{x}$  and  $\vec{y}$  are unit direction vectors in cartesian coordinates.

The electric field is shown in Figure 4(b). Another electric component  $\delta E_F / \delta \epsilon_{12} \nabla \epsilon_{12}$  ( $\epsilon_{12}$  is shear strain) resulting from shear strain may be much smaller than main components in Equation (7) for an isotropic material and is neglected here, nevertheless, for other materials this component must be considered. By means of simple estimations, the electric field near the core of edge dislocation may approach a value as high as  $10^8$  V/m. Note that the electric field and potential strongly rest with  $V_0 \delta E_F / \delta V$ , an intrinsic



feature of a material. For sodium chloride, if  $V_0 \delta E_F / \delta V$  is positive, the dislocation may behave as an entirety of positive charges in tensile regions and negative charges in compressed regions, and their charge densities are highest at the core as is shown in Figure 4(a). For instance, for a NaCl bar plastically bent by dislocations under tensile load, production and migration of the dislocations may generate a tensile stress field at convex surface and a compressed stress field at concave surface, thereby creating positive charges at the convex surface and negative charges at the concave surfaces, which is indeed the experimental observations [8].

Another type of widespread dislocations, screw dislocations, only induce shear strains in an isotropic material [13]. Using Equation (1) in the same way, the induced electric field, potential difference and electric energy around a screw dislocation may be obtained as followings,

$$\vec{E}(x, y) = \frac{1}{q} \frac{\delta E_F}{\delta \epsilon_{13}} \frac{b_s}{4\pi} \left[ \frac{-x^2 + y^2 + 2xy}{(x^2 + y^2)^2} \vec{x} + \frac{-x^2 + y^2 - 2xy}{(x^2 + y^2)^2} \vec{y} \right] \quad (9)$$

$$V_Z(x_2, y_2) - V_Z(x_1, y_1) \approx -\frac{1}{q} \frac{\delta E_F}{\delta \epsilon_{13}} \frac{b_s}{4\pi} \left( \frac{x_2 - y_2}{x_2^2 + y_2^2} - \frac{x_1 - y_1}{x_1^2 + y_1^2} \right) \quad (10)$$

where  $b_s$  is Burgers vector of this screw dislocation in  $z$  direction,  $\epsilon_{13}$  is induced shear strain by this screw dislocation. In isotropic materials, this relation may exist  $|\delta E_F / \delta \epsilon_{13}| \ll |\delta E_F / \delta \epsilon_{11}|$  and there is no electric charges at its core. Therefore, *Yuheng Zhang effect* and its electric field around a screw dislocation may be not as apparent as that of edge dislocations, as is the point observed by experiments [8].

The electric field around dislocations necessarily brings electric energy. For an isotropic linear dielectric material, the electric energy for a dislocation of unit length

may be easily obtained in terms of Equation (7) and (9),

$$G_e \approx \frac{\varepsilon_0 \varepsilon}{8\pi} \left( \frac{V_0}{q} \frac{\delta E_F}{\delta V} \right)^2 \frac{(1-2\nu)^2 b_e^2}{(1-\nu)^2 r_{e0}^2} \quad (11)$$

$$G_s \approx \frac{\varepsilon_0 \varepsilon}{16\pi} \left( \frac{1}{q} \frac{\delta E_F}{\delta \mathcal{C}_{13}} \right)^2 \frac{b_s^2}{r_{s0}^2} \quad (12)$$

where  $\varepsilon$  is relative dielectric permittivity,  $G_e$ ,  $G_s$  is electric energy surrounding an edge dislocation and a screw dislocation, respectively,  $r_{e0}$ ,  $r_{s0}$  is radius of the corresponding edge dislocation core and screw dislocation core. For some dielectric materials, this energy may be comparable with or even larger than the elastic energy of dislocations.

In the case of metals, the calculation of electric energy may be a little complicated due to the difficult selection of relative permittivity. A potential candidate may reside in Thomas-Fermi screening dielectric function, i.e., the static limit of Lindhard dielectric function. Take the same approximation as in calculations of the earth's geomagnetic field [2] and it is

$$\varepsilon_0 (e^{r/r_s} - 1) E = n_e q r \approx C \quad (13)$$

where  $r_s$  is Thomas-Fermi screening length,  $n_e$  denotes free electron density in metals,  $q$  stands for electron charge and  $E$  is electric field strength. For dislocation inducing static electric field in most of metals,  $C$  may approximate  $\sim 10 \text{ C/m}^2$ . Thus, electric energies around an edge dislocation and a screw dislocation is obtained,

$$G_e \approx \left| \frac{C}{q} \frac{V_0 \delta E_F}{\delta V} \frac{(1-2\nu) b_e}{(1-\nu)} \right| \ln \frac{R}{r_{e0}} \quad (14)$$

$$G_s \approx \left| \frac{C}{q} \frac{\delta E_F}{\delta \mathcal{C}_{13}} \frac{\sqrt{2} b_s}{2} \right| \ln \frac{R}{r_{s0}} \quad (15)$$

where  $R$  is radius of strained regions. The calculation only give rough estimation of

electric energies of dislocations in metals, and more accurate calculations still need to be explored.

According to above discussions, total energy of a dislocation may consist of two parts, one is the well-known elastic energy, the other is the electric energy which may be pointed out for the first time here. When analyzing dislocation dominated mechanical properties of materials, one must take this electric energy into account. Due to electrostriction effect, the electric field reversely leads to strains which may modify strain field of dislocations. In the other respect, the dislocations may affect electrical resistance of materials. In view of the electric fields of edge and screw dislocations (Figure 4(b)), electrical resistance in  $x$ - $y$  plane may increase conspicuously as is usually observed in metals and semiconductors [14, 15, 16], but electrical resistance along  $z$  direction may not be influenced due to absence of  $z$ -component electric field. This mechanism provides people a simple route to understand dislocation effect on electrical resistance. For isotropic materials, main strain induced lift of  $E_F$  may be much larger than shear strain induced lift, therefore, the edge dislocations may dominate variations of electrical resistance. For anisotropic materials, one must consider influence of both the two types of dislocations on electrical resistance.

Moreover, edge dislocation-constructed small-angle-tilt grain boundaries is anticipated to exhibit charge separation at convex and concave surfaces and the charge sign rests with strain dependence of Fermi surface energy. Furthermore, electric effects of other mechanical microstructures composed of dislocations could be clarified by means of corresponding strain field.

## ii) Other point defects

Analogous to the case of dislocations, beside elastic stress field around them, other point defects such as interstitial atoms, vacancies, impurity atoms and so on usually give rise to radial strain field around them. In light of *Yuheng Zhang effect*, the radial strain may generate radial electric field. Using Gauss's law, one may find that the point defects act as charged particles in materials and increase electrical resistance in all crystalline orientations. When analyzing multi-physical properties of materials, one must consider both elastic field and related radial electric field.

## iii) Abrikosov vortex

In type-II superconductors, when the external magnetic field is higher than critical field  $H_{C1}$ , magnetic field will enter the superconductor and forms some vortices. For an isolated vortex, density of Cooper pairs is suppressed to zero at its center and restores its full value at a distance of coherence length  $\xi$  away from the center [17]. Within the vortex, materials stay in magnetic field and its magnetic pressure coming from Lorenz force may be  $P(r) = B^2(r)/2\mu_0$  [18], where  $P(r)$  is magnetic pressure,  $r$  is distance from the center of vortex,  $\mu_0=4\pi\times 10^{-7} H/m$  is vacuum magnetic susceptibility. At the center of vortex, superconducting state is believed to be completely destroyed, and the corresponding magnetic field might reach upper critical field, *i.e.*,  $B(0)=B_{C2}(T)$ . So, the generated tensile magnetic pressure at the center is

$$P(0) = \frac{B_{C2}^2(T)}{2\mu_0} \quad (16)$$

For type-II superconductors, the upper critical field varies as temperature and is usually of order  $\sim 10 T$  at low temperatures [19, 20, 21]. The generated tensile magnetic

pressure may be  $P(0) \approx 40$  MPa, comparable with yield strength of many metals, and the induced volume strain may reach  $\sim 10^{-4}$ . Based on *Yuheng Zhang equation*, if magnitude  $\delta E_F / \delta \epsilon$  ranges from 1 to 10 eV, the produced potential  $V_Z(T)$  will be in the range 0.1~1 mV. On the other hand, strong magnetic field may also alter the Fermi surface energy of this normal region a little by  $\mu_B \cdot B_{C2}(T) \sim 0.6$  meV, producing another potential. So, the whole electrical potential between center of vortex and its surrounding areas may be summation of the two potentials. Like temperature dependence of upper critical field  $B_{C2}(T)$ , the potential is strongly temperature dependent and decreases rapidly as temperature increases. Conversely, measurement of temperature dependence of electrical potential  $V_Z(T)$  for a single vortex may offer people another method to determine temperature dependence of upper critical field  $B_{C2}(T)$ , particularly upper critical field of some high-temperature superconductors such as cuprate superconductors and iron-based superconductors. Owing to the electric properties, Abrikosov vortices are expected to behave as line charges sometimes.

#### **iv) Magnetic domain walls and magnetic skyrmions**

When a ferromagnetic (FM) material is subject to external magnetic field, Fermi surface energy of magnetic domains parallel to magnetic field may be lower than that of domains antiparallel to the magnetic field. In case of popular  $180^\circ$  domain walls, Fermi surface energy may be lifted by  $2S\mu_B B$  across the domain wall, where  $S\mu_B$  is magnetic moment at a site,  $\mu_B$  is Bohr magneton,  $B$  is external magnetic field. So according to *Yuheng Zhang equation*, an electric field and an electrical potential may appear within the domain wall. Assume 1 T of the external magnetic field,  $2\mu_B$  of the

magnetic moment, and 30 nm of the domain wall thickness, the corresponding electrical potential and related electric field may approach  $\sim 0.2$  mV and  $\sim 7 \times 10^3$  V/m, respectively.

Like the case of magnetic domain walls, a recent hot and interesting topic of a topological defect, magnetic skyrmion, may display similar electric properties under external field and may yield an electrical potential and a radial electric field between center of skyrmion and its surroundings. In a sense, skyrmions also behave as charged quasiparticles under external magnetic field, which means that it could be tuned by an external electric field.

#### **(5) Phase transition induced electrical voltage at interface**

As discussed for metals [1], when phase transition happens for a material, Fermi surface energy of the new phase may be lifted due to volume and symmetry changes. So, an electrical voltage and electric polarization may appear at the interface between the material and its new phase in spite of their identical chemical compositions. This voltage may be written as  $qV_{12} = E_{F2} - E_{F1}$ , where  $V_{12}$  is voltage at the interface,  $q$  is electron charge,  $E_{F1}$  and  $E_{F2}$  are the Fermi surface energy of the material and its new phase, respectively. This voltage may be unique and likes a fingerprint, and can be utilized to characterize phase transitions, especially the liquid-liquid transitions which are challenging for ordinary diagnostic methods.

#### **(6) Strain gradient induced electric polarization**

Upon strain gradient, insulators may present an electric polarization and it is obtained by means of *Yuheng Zhang equation*,

$$\vec{P}(\vec{r}) = \varepsilon_0 (\varepsilon - 1) \frac{\nabla E_F(\vec{r})}{q} \quad (17)$$

where  $\vec{P}(\vec{r})$  is position dependence of electric polarization,  $\epsilon_0$  and  $\epsilon$  are vacuum dielectric permittivity and relative permittivity, respectively. Obviously, for a given material, the electric polarization depends on gradient of Fermi surface energy, and a larger position gradient of Fermi surface energy would lead to a larger electric polarization. For instance, the electric polarization may be very large within rarefaction waves and shock wave front which will be discussed later.

### **(7) Mechanical fracture of material**

For cases of mechanical loading exceeding fracture strength of a solid material, a fracture appears and a crack propagates in the material and stress concentration usually occurs at the tip of crack, especially for some ductile metals. The crack propagation is usually accompanied by elastic strain energy release and strain release at its both sides, including initial concentrated crack tip stress and strain. The dynamic mechanical processes may give rise to *Yuheng Zhang effect*, thereby creating an electric field and electrical potential along unloading directions. From the viewpoint of energy, when a crack starts and propagates, the elastic strain energy transforms to fracture energy and electric energy, despite much smaller value of the electric energy than fracture energy. So the strain induced polarization state may help people examine strain state of a material under loading.

### **(8) Gravity induced electric field in planets**

Some materials naturally exhibit strain gradient due to gravity, *e.g.* various types of rocks in some planets and satellites in the aerospace. Generally speaking, a rock stays in planets deeper, it is subject to a larger pressure and thereby displays a larger

compressed strain, resulting in a radial electric field and a corresponding electric polarization. The field strength may reach an order of  $\sim 10^{-6}$  V/m. However, the actual electric field may be relatively complicated because of many different types of rocks and minerals as well as their complex distributions in planets and satellites.

Information of electric field distributions of minerals in the earth may be employed to unravel evolution of their stress and strain state. Of specially pointed out is the possibility that in terms of scanning internal electric field of minerals in the earth's crust, earthquake forecast, a highly desired but great challenge for people, might be viable in the future.

#### **(9) Microscopic origin of flexoelectric effect**

Flexoelectric effect was stated by phenomenological analysis more than half a century ago and it is believed that strain gradient has directionality and can yield an electric polarization in dielectric materials [22]. The flexoelectric polarization is given by the following relation [12],

$$P_l = \mu_{ijkl} \frac{\partial \epsilon_{ij}}{\partial x_k} \quad (18)$$

where  $P_l$  is the  $l$ -th component of flexoelectric polarization,  $\mu_{ijkl}$  is flexoelectric coefficient and a fourth-rank polar tensor,  $\epsilon_{ij}$  is position dependence of strain and  $x_k$  is position coordinate. Previous theoretical analysis [22, 23] estimated that magnitude of flexoelectric coefficient was very small and was the order of  $q/a \sim 10^{-10}$  C/m ( $q$  is electron charge and  $a$  is lattice parameter), meaning that flexoelectric effect might be too small to tune polarization of materials. However, it was later found that flexoelectric coefficient can reach an unexpected large value and the related flexoelectric effect can



dominate electric polarization of a material. For an instance, magnitude of flexoelectric coefficient is order of  $10^{-6}$  C/m in relaxor ceramics  $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3$  [24], several orders higher than formerly estimated; and stress gradient can mechanically switch polarization in ferroelectric film [25]. Since then many progresses have been made in this area, but the underlying microscopic physical origin is still a puzzle.

Here, it is stated and believed that flexoelectric effect is a special case of *Yuheng Zhang effect* and *Yuheng Zhang potential* in dielectric materials. And the microscopic origin may be provided by *Yuheng Zhang equation*, i.e., Equation (1). So, the induced electric polarization is given

$$P_l = \frac{\varepsilon_0(\varepsilon-1)}{q} \frac{\delta E_F(\vec{r})}{\delta \mathcal{C}_{ij}(\vec{r})} \frac{\partial x_k}{\partial x_l} \frac{\partial \mathcal{C}_{ij}(\vec{r})}{\partial x_k} \quad (19)$$

where  $\mathcal{C}_{ij}(r)$  is position dependence of strain,  $q$  is electron charge. By comparing Equation (18) and (19), the flexoelectric coefficients can be obtained easily,

$$\mu_{ijkl} = \frac{\varepsilon_0(\varepsilon-1)}{q} \frac{\delta E_F(\vec{r})}{\delta \varepsilon_{ij}(\vec{r})} \delta_{kl} \quad (20)$$

where  $\delta_{kl}$  is unity only when the subscript is the same as each other ( $\delta_{kl}=1$  for  $k=l$ ;  $\delta_{kl}=0$  for others). So, the flexoelectric coefficients can be reduced and be expressed as,

$$P_l = \mu_{ij} \frac{\partial \mathcal{C}_{ij}(\vec{r})}{\partial x_l} \quad (21)$$

$$\mu_{ij} = \frac{\varepsilon_0(\varepsilon-1)}{q} \frac{\delta E_F(\vec{r})}{\delta \mathcal{C}_{ij}(\vec{r})} \quad (22)$$

According to Equation (22), flexoelectric coefficients are monitored by two key parameters: one is the relative permittivity  $\varepsilon$ , the other is strain dependence of Fermi surface energy  $\delta E_F(\vec{r})/\delta \mathcal{C}_{ij}(\vec{r})$ . For most of materials,  $|\delta E_F(\vec{r})/\delta \mathcal{C}_{ij}(\vec{r})|$  may be in the range 1~10 eV, thus the magnitude of corresponding flexoelectric coefficients are

summarized in Table 1. In the case of relaxor  $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3$  whose relative permittivity is 13000 at 1  $\text{KHz}$  and room temperature, flexoelectric experimental measurement gives the corresponding flexoelectric coefficient  $4 \times 10^{-6} \text{ C/m}$  [24], which is in qualitative agreement with estimation in Table 1. For epitaxial single crystalline  $\text{BaTiO}_3$  thin film under compression, based on its calculated volume variations [25], the electric field estimated by *Yuheng Zhang equation* gives  $5 \times 10^7 \sim 5 \times 10^8 \text{ V/m}$ , which is in accord with the measured biased field  $\sim 10^8 \text{ V/m}$  in the hysteresis loop in supplementary materials [25]. Further quantitative work may need to carry out first-principle calculations to determine the parameter  $\delta E_F(\vec{r})/\delta C_{ij}(\vec{r})$  of these materials.

#### **(10) Strain engineering**

In thin films epitaxially grown on substrate, strains usually stem from lattice mismatch between the film material and substrate. For the mismatch induced strains, there exist a critical thickness within which the strains offer an opportunity to enhance some physical properties of the film and beyond which the strains are not available [26]. The out-of-plane strain can be expressed by the following form [26],

$$C(r) = C_0 e^{-r/d_c} \quad (23)$$

where  $C(r)$  stands for out-of-plane strain in the film at a distance  $r$  away from the interface,  $d_c$  is the related critical thickness, and  $C_0$  is the out-of-plane strain at the interface. The critical thickness is usually tens of nanometers, and strain gradient in oxide epitaxial thin films such as  $\text{HoMnO}_3$  can often approach a value as large as  $10^6 / \text{m}$  [27]. Combining *Yuheng Zhang equation* and Equation (23), one may easily find that the generated internal electric field is usually perpendicular to the interface. It may be

the physical origin for the self-polarized ferroelectric domain, i.e., upward or downward polarization shown in thin films in their as-grown state [27].

To ascertain whether the proposed point is right or not, the corresponding comparison is performed between the theory and experiments. In terms of *Yuheng Zhang equation*, if the corresponding parameter  $\delta E_F(\vec{r})/\delta C_{ii}(\vec{r})$  for HoMnO<sub>3</sub> is order of  $\sim 10$  eV, the yielded intrinsic electric field might be  $10^7$  V/m agreeing with experimental biased electric field  $\sim 4 \times 10^7$  V/m measured in the hysteresis loop for the ferroelectric monodomain [28]. Investigations on strained SrTiO<sub>3</sub> epitaxial thin film suggest that the mismatch strain play a similar role as an electric field in polarization [29]. For simplicity, as an estimation, the strain gradient could be taken as ratio between strain and film thickness  $2 \times 10^5$  /m and therefore the induced electric field may achieve  $10^6$  V/m, a value consistent with the biased field  $\sim 2 \times 10^6$  V/m in measured hysteresis loop [29]. In a word, these experiments may verify correctness of *Yuheng Zhang equation* and *Yuheng Zhang effect*, albeit qualitative nowadays.

### **(11) Microscopic theory of shock polarization**

For shock-loaded materials as shown in Figure 1, *Yuheng Zhang potential* and its corresponding electric polarization always exist across the shock wave front (region II in Figure 1) no matter how weak the shock wave is and no matter whether plasticity occurs or not. Based on *Yuheng Zhang equation* and *Yuheng Zhang potential*, magnitude of this potential and electric polarization relies on shock waves. A stronger shock wave may lead to a larger *Yuheng Zhang potential* and a larger electric polarization. In reality, the series of phenomena may be first discovered by A. V. Stepanov [30] and were

established by subsequent large amount of experimental work that upon shock induced elastic and plastic deformations almost all types of crystals and even some liquid such as liquid water give rise to an electric polarization and a related electrical voltage, namely, shock polarization [31, 32, 33, 34, 35].

As for shock polarization, many people attempted to interpret it and put forward many mechanisms for different materials, such as microscopic linear ionic chain model [36] and dislocation model [37] for ionic crystals, phenomenological mechanical polarization model for plastics [38], polar molecule rotating model for polar dielectrics [39] and so on. However, the microscopic mechanism for shock polarization is still little known and is challenging.

In this work, it is proposed and stated that shock polarization is another representation of *Yuheng Zhang effect* and *Yuheng Zhang potential* in materials under shock loading. The corresponding microscopic mechanism is given by *Yuheng Zhang equation*.

Figure 5 (a) gives the typical experimental setup for the measurements [33, 37, 40], and its equivalent electrical circuit is shown in Figure 5 (b) and (c). On imparting a screening metal plate (Electrode I) by a flyer plate, shock waves emerge and compress the metal electrode, thereby creating a *Yuheng Zhang potential*  $V_{S1}$  in this electrode (left brown zones in Electrode I). Before shock wave reaching interface between Electrode I and the investigated materials, its equivalent electrical circuit is given by Figure 5 (b). As a result, there is no electrical signals, which is consistent with experimental results [33, 37, 40].

### (i) Electromotive force (EMF) in shocked metals

When shock waves arrive at the boundary between Electrode I and investigated metal, they enter the metal and compress it (red zones), generating a potential  $V_m$  between shocked regions and un-shocked regions as shown in Figure 5(a). Simultaneously another wave is reflected into Electrode I, compressing or decompressing Electrode I further (If shock impedance of investigated material is larger than that of Electrode I, the reflected wave is a shock wave that compresses Electrode I, otherwise it is rarefaction waves decompressing the electrode.). Therefore, another potential  $V_{S2}$  is yielded in Electrode I (magenta zones) as shown in Figure 5(a). The equivalent electrical circuit is given by Figure 5(c). If the investigated metal is the same as electrode metal, no wave is reflected from the interface, and the potentials  $V_{S2}$ ,  $V_m$  in circuit (Figure 5(c)) equals to the potential  $V_{S1}$ , *i.e.*,  $V_{S2} = V_m = V_{S1}$ . Therefore, there is no electrical signals, which is in agreement with experimental results [41].

Now let us focus on the case that the electrodes and investigated sample are different metals. All the formed plane boundaries between electrodes and sample could be regarded as good electrical contacts in the circuit shown in Figure 5(c). In the circuit, a resistance  $R_2$  denotes intrinsic resistance of metal sample, electrodes and their contact resistance. A resistance  $R_1$  stands for contact resistance between electrode I and released lateral regions in the metal sample (light blue regions in Figure 5(a)), the contact area may be a very narrow ring. The two resistances may be much smaller than the external resistance  $R_0$  which is in the range 50~100  $\Omega$  in the measurements [33, 37, 40, 42]. So the measured voltage might be

$$V_0 = \frac{R_1 (V_{S2} - V_m)}{R_1 + R_2} \quad (24)$$

$$V_{S2} = E_{FS} \left[ (1 + \epsilon_{S2})^{-2/3} - 1 \right] \quad (25)$$

$$V_m = E_{Fm} \left[ (1 + \epsilon_m)^{-2/3} - 1 \right] \quad (26)$$

where  $V_0$  is the measured voltage,  $V_{S2}$ ,  $V_m$  and  $E_{FS}$ ,  $E_{Fm}$  are electrical potential and Fermi surface energy of screening metal plate and sample, respectively,  $\epsilon_{S2}$ ,  $\epsilon_m$  is related strains of screening metal plate and sample (negative for compressed strain and positive for tensile strain ).

Due to uncertainty in the experiments, the accurate values of the two resistances is difficult to determine, which may be the reason for conspicuous discrepancy of the measured voltage in different groups [33, 42, 43, 44, 45]. However, it can give the sign of the measured voltage. If the potentials fulfill  $V_{S2} - V_m > 0$ , the measured voltage is positive, and conversely it is negative. If a metal possessing both high Fermi surface energy and low shock impedance is used as screening metal plate, in other words, a large positive potential  $V_{S2}$ , the measured voltage seems more to be positive, otherwise it is more possible to be negative, which agrees with experimental results [33]. Interestingly, according to Equation (24), if an insulating material whose shock impedance matches that of sample is placed around the sample, *i.e.*, increasing  $R_1$  dramatically, the measured voltage would increase and be close to the value  $(V_{S2} - V_m)$ . This point still needs experiments to prove in the future.

Of noted is that in the above analysis on EMF in metals, the thermoelectric power and phase transition are not considered. Both of them can contribute to the experimental results, especially the latter one. To be anticipated, when phase transition happens, it

usually causes a sudden magnitude change or a sign change of the measured voltage. The precise value of phase-transition-induced electrical voltage still need much theoretical and experimental work to determine.

## (ii) EMF in shocked semiconductors and insulators-shock polarization

As for most of semiconductors and insulators, their resistivity is usually larger than  $10^6 \Omega \cdot \text{m}$ . So for a sample with a thickness  $\sim \text{mm}$  and an area  $\sim \text{cm}^2$ , its resistance is much larger than the external resistance  $R_0$  in a range 50-100  $\Omega$ . If the resistance of the sample in compressed regions does not decrease sharply and is still much larger than external resistance  $R_0$ , the related electrical circuit could be simplified as shown in Figure 5(d), which is analogous to a previous phenomenological explanation for shock polarization [46]. The shocked regions and unshocked regions of sample may act as two parallel-plate capacitors with relative permittivity  $\epsilon_3$ , electric field  $E_3$ , capacitance  $C_3$  and  $\epsilon_1$ ,  $E_1$ ,  $C_1$ , respectively. The time dependence of measured electrical current and voltage are given as

$$I(t) = \frac{dC_3 V_3}{dt} \quad (27)$$

$$I(t) = \frac{dC_1 V_1}{dt} \quad (28)$$

$$I(t) R_0 = V_{s2} - V_m - V_{\frac{3}{3}} V \quad (29)$$

where  $I(t)$  is time dependence of electrical current passing the circuit (positive for clockwise),  $V_3$  and  $V_1$  are the voltages at the two capacitors, respectively. After some simple calculations, it is

$$\frac{dE_1(t')}{dt'} + [(\lambda_3 - \lambda_1)t' + \lambda_1] E_1(t') = \lambda_1 E_0 \quad (30)$$

where the parameters  $\lambda_1=t_0/\tau_1$ ,  $\lambda_3=t_0/\tau_3$ ,  $\tau_1=A\varepsilon_0\varepsilon_1R_0/L$ ,  $\tau_3=A\varepsilon_0\varepsilon_3R_0/(U-u)t_0$ ,  $t'=t/t_0$ ,  $V_1=U(t_0-t)E_1(t)$ ,  $V_3=(U-u)tE_3(t)$ ,  $L=Ut_0$ ,  $E_0=(V_{S2}-V_m)/L$ ,  $U$  is shock wave velocity and  $u$  is the particle velocity,  $L$  is thickness of the sample,  $A$  is area of sample,  $t_0$  is time duration for shock waves passing the sample,  $E_1$  is electric field in capacitor  $C_1$ . Based these equations, the measured time dependence of electrical current is

$$I(t') = \frac{V_{S2}-V_m}{R_0} \left\{ 1 - \sqrt{\frac{\pi\lambda_3(k-1)}{2}} \left( \frac{k}{k-1} - t' \right) e^{\frac{\lambda_3(k-1)}{2} \left( t' - \frac{k}{k-1} \right)^2} \left[ \operatorname{Erf} \left( \sqrt{\frac{\lambda_3}{2}} \frac{k}{\sqrt{k-1}} \right) + \operatorname{Erf} \left( \sqrt{\frac{\lambda_3(k-1)}{2}} \left( t' - \frac{k}{k-1} \right) \right) \right] \right\} \quad (31)$$

where parameter  $k=\lambda_1/\lambda_3$ ,  $\operatorname{Erf}(x)$  is error function. Seen from this equation, the current versus time relation  $I(t')$  only depends on parameters  $k$  and  $\lambda_3$ . To clarify roles of sample area, thickness and relative permittivity and so on, let sample area  $A=\underline{A} \times 10^{-4} \text{ m}^2$ , external resistance  $R_0=\underline{R} \times 50 \text{ } \Omega$ ,  $V_{S2}-V_m=\underline{V} \times 1 \text{ V}$ , velocity  $U-u=\underline{U} \times 10^3 \text{ m/s}$ , permittivity  $\varepsilon_0\varepsilon=\underline{\varepsilon} \times 10^{-11} \text{ F/m}$  and duration time  $t_0=\underline{t} \times 10^{-6} \text{ s}$ , where  $\underline{A}$ ,  $\underline{R}$ ,  $\underline{U}$ ,  $\underline{\varepsilon}$  and  $\underline{t}$  are dimensionless parameters. Substitute them into Equation (31), it is

$$I(t') = \frac{0.02V}{\underline{R}} \left\{ 1 - 10^2 \underline{t} \sqrt{\pi(k-1)} \sqrt{\frac{\underline{U}}{A\underline{\varepsilon}\underline{R}}} \left( \frac{k}{k-1} - t' \right) e^{\frac{10^4 \underline{U}^2}{A\underline{\varepsilon}\underline{R}} (k-1) \left( t' - \frac{k}{k-1} \right)^2} \left[ \operatorname{Erf} \left( 10^2 \underline{t} \sqrt{\frac{\underline{U}}{A\underline{\varepsilon}\underline{R}}} \frac{k}{\sqrt{k-1}} \right) + \operatorname{Erf} \left( 10^2 \underline{t} \sqrt{\frac{(k-1)\underline{U}}{A\underline{\varepsilon}\underline{R}}} \left( t' - \frac{k}{k-1} \right) \right) \right] \right\}$$

According to this equation, the typical results, current-time relation  $I(t')-t'$  are shown in Figure 6. It shows that  $I(t')-t'$  line presents a down-concave shape (see Figure 6(a)), and interestingly, magnitude of the current relies on both sample areas and duration times, in other words, sample thickness. The electrical current gradually increases with increasing sample area, and it remarkably rises as sample thickness declines (see Figure 5(b)), which are consistent with experimental results for single crystal sodium chloride NaCl [37, 40]. Through analysis, it is found that these properties arise from the pronounced increase of relative permittivity of materials in



compressed regions. To one's surprise, if the relative permittivity decreases after shock compression, the current decreases as time and would display a sign change as shown in Figure 6(c), which may be reasons for the observed sign change for shock-loaded single crystal NaCl at 1 and 1.6 GPa [37, 40]. In reality, the sign of current primarily depends upon variations of electrical potentials  $V_{S2}$  of the metal screening plate and  $V_m$  of the sample under compression. As discussed previously, the current is positive in the case of  $V_{S2}-V_m>0$ , but is negative in the opposite case. Furthermore, another  $I(t')$ -  $t'$  relation is predicted as shown in Figure 6 (d), which needs to be proved in the future. These primary results are stable in broad parameter zones. And the quantitative comparison with experiments nevertheless need first-principle calculations to determine potential values  $V_{S2}$  and  $V_m$ .

In terms of above discussions, shock polarization, a widespread phenomenon in materials under shock loading, may be fundamentally understood by using *Yuheng Zhang effect* and *Yuheng Zhang potential*. However, of emphasized is that phase transition, piezoelectric effect and self-polarization such as ferroelectric polarization and so on are not taken into account here. When one investigates a specific material under shock loading, he (she) must consider all possible factors.

## **Conclusion**

In summary, in this work we point out that Fermi surface energy of a material is position dependence, and accordingly uncover a new physical law as well as a physical effect in condensed materials, i.e, *Yuheng Zhang equation* and *Yuheng Zhang effect*. By investigations using this law and this effect, we gain new understandings of multi-

physical properties of materials: 1) materials experiencing strain gradient could present current-rectifying properties; 2) thermal expansion (contraction) and *Yuheng Zhang effect* contribute to thermoelectric power and Seebeck coefficient obviously through *Yuheng Zhang coefficient*; 3) *Yuheng Zhang effect* gives a new mechanism of electron-phonon interaction; 4) some crystalline defects and topological defects in materials, such as dislocations, Abrikosov vortices and so on, are often accompanied by a surrounding electric field and an electrical potential; 5) strain gradient can cause an electric polarization in dielectric materials; 6) the gravity induced electric field persists in the earth, which may be useful for earthquake research; 7) microscopic physical origin of flexoelectric effect, a challenge for more than half a century in the world, may be figured out by *Yuheng Zhang effect*, and the flexoelectric coefficients can be derived directly using *Yuheng Zhang equation*; 8) microscopic theory of shock polarization, another long-standing problem in the world, may be clarified in terms of *Yuheng Zhang effect*. In all, *Yuheng Zhang effect* offer people new understanding of electric properties of strained materials, and may find many application in multi-areas.

## Reference:

- [1] Yuanjie Huang, Yuheng Zhang Effect: Strain-induced electric effect in metals, *viXra*: 1707.0147 (2017).
- [2] Yuanjie Huang, Possible source of the earth's geomagnetic field, *viXra*: 1709.0024 (2017).
- [3] Holler, F. James; Skoog, Douglas A.; Crouch, Stanley R. Principles of Instrumental Analysis (6th ed.). Cengage Learning. (2007). p. 9.
- [4] IUPAC Gold Book, definition of Galvani potential difference.
- [5] Philip Phillips, Advanced Solid State Physics [b], Westview Press (USA), (2003), p.179.
- [6] G. Gruner, The dynamics of charge-density waves, *Rev. Mod. Phys.* **60**, No.4, 1129-1181(1988).
- [7] A. Hikata, C. Elbaum, B. Chick, R. Truell, Electrical-charge study in sodium chloride during plastic deformation, *J. Appl. Phys.* **34**, No.8, 2154-2158(1963).
- [8] Frederick Seitz, Speculations on the properties of the silver halide crystals, *Rev. Mod. Phys.* **23**, No.4, 328-353(1951)
- [9] R. J. Schwensfeir, C. Elbaum, Electric charge on dislocation arrays in sodium chloride, *J. Phys. Chem. Solids*, **28**, 597-606(1967).
- [10] V. Nagarajan, C. L. Jia, H. Kohlstedt, R. Waser, I. B. Misirlioglu, S. P. Alpay, R. Ramesh, Misfit dislocations in nanoscale ferroelectric heterostructures, *Appl. Phys. Lett.* **86**, 192910(1-3)(2005).
- [11] S. P. Alpay, I. B. Misirlioglu, V. Nagarajan, R. Ramesh, Can interface dislocations

- degrade ferroelectric properties? *Appl. Phys. Lett.* **85**, No.11, 2044-2046 (2004).
- [12] Daesu Lee, Tae Won Noh, Giant flexoelectric effect through interfacial strain relaxation, *Phil. Trans. R. Soc. A* **370**, 4944–4957(2012).
- [13] Ulrich Messerschmidt, Dislocation dynamics during plastic deformation (Springer Series in Materials Science 129), Springer, (2010), p.42.
- [14] K. X. Chen, Q. Dai, W. Lee, J. K. Kim, E. F. Schubert, J. Grandusky, M. Mendrick, X. Li, and J. A. Smart, Effect of dislocations on electrical and optical properties of n-type  $\text{Al}_{0.34}\text{Ga}_{0.66}\text{N}$ , *Appl. Phys. Lett.* **93**, 192108 (2008).
- [15] A. A. Allerman, M. H. Crawford, A. J. Fischer, K. H. A. Bogart, S. R. Lee, D. M. Follstaedt, P. P. Provencio, and D. D. Koleske, Growth and design of deep-UV (240–290nm) light emitting diodes using AlGa<sub>N</sub> alloys, *J. Cryst. Growth* **272**, 227 (2004).
- [16] Shakti Chauhan and Ashraf F. Bastawros, Probing thickness-dependent dislocation storage in freestanding Cu films using residual electrical resistivity, *Appl. Phys. Lett.* **93**, 041901 (2008).
- [17] R. P. Huebener, Dynamics of magnetic flux structures in superconductor, *Physics Reports* (Section C of Physics Letters), **13**, No. 4, 143-189(1974).
- [18] [https://en.wikipedia.org/wiki/Magnetic\\_pressure](https://en.wikipedia.org/wiki/Magnetic_pressure)
- [19] H. H. Wen, S. L. Li, Z. W. Zhao, H. Jin, Y. M. Ni, Z. A. Ren, G. C. Che and Z. X. Zhao, Magnetic relaxation and critical current density of the new superconductor  $\text{MgB}_2$ , *Supercond. Sci. Technol.* **15**, 315–319 (2002).
- [20] Haihu Wen, Flux dynamics and vortex phase diagram of cuprate superconductors, *Physics*, 35, No.1, 16(2006)

- [21] C. S. Yadav, P. L. Paulose, Upper critical field, lower critical field and critical current density of  $\text{FeTe}_{0.60}\text{Se}_{0.40}$  single crystals, *New Journal of Physics* **11**, 103046(2009).
- [22] Sh. M. Kogan, Piezoelectric effect during inhomogeneous deformation and acoustic scattering of carriers in crystals, *Sov. Phys. Solid State* **5**, 2069 (1964).
- [23] A. K. Tagantsev, Piezoelectricity and flexoelectricity in crystalline dielectrics. *Phys. Rev. B* **34**, 5883-5889(1986).
- [24] Wenhui Ma and L. Eric Cross, Observation of the flexoelectric effect in relaxor  $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3$  ceramics, *Appl. Phys. Lett.* **78**, 2920 (2001).
- [25] H. Lu, C.-W. Bark, D. Esque de los Ojos, J. Alcala, C. B. Eom, G. Catalan, and A. Gruverman, Mechanical writing of ferroelectric polarization, *Science*, **336**, 59-61(2012).
- [26] H. Joon Kim, S. Hoon Oh, and Hyun M. Jang, Thermodynamic theory of stress distribution in epitaxial  $\text{Pb}(\text{Zr}, \text{Ti})\text{O}_3$  thin films, *Appl. Phys. Lett.* **75**, 3195 (1999).
- [27] S. H. Baek, H.W. Jang, C. M. Folkman, Y. L. Li, B. Winchester, J. X. Zhang, Q. He, Y. H. Chu, C. T. Nelson, M. S. Rzchowski, X. Q. Pan, R. Ramesh, L. Q. Chen and C. B. Eom, Ferroelastic switching for nanoscale non-volatile magnetoelectric devices, *Nature Materials*, **9**, 309(2010).
- [28] D. Lee, A. Yoon, S.Y. Jang, J.-G. Yoon, J.-S. Chung, M. Kim, J. F. Scott, and T.W. Noh, Giant flexoelectric effect in ferroelectric epitaxial thin films, *Phys. Rev. Lett.* **107**, 057602 (2011).
- [29] H.W. Jang, A. Kumar, S. Denev, M. D. Biegalski, P. Maksymovych, C.W. Bark, C. T. Nelson, C. M. Folkman, S. H. Baek, N. Balke, C. M. Brooks, D. A. Tenne, D. G.

- Schlom, L. Q. Chen, X. Q. Pan, S.V. Kalinin, V. Gopalan, and C. B. Eom, Ferroelectricity in strain-free SrTiO<sub>3</sub> thin films, *Phys. Rev. Lett.* **104**, 197601 (2010).
- [30] A. W. Stepanow, *Zs. Phys.* **81**, 560 (1933).
- [31] J. E. Caffin and T. L. Goodfellow, Electrical effects associated with the mechanical deformation of single crystals of alkali halides, *Nature*, **176**, 878 (1955).
- [32] G. E. Hauver, Shock-Induced Polarization in Plastics. II. Experimental Study of Plexiglas and Polystyrene, *J. Appl. Phys.* **36**, 2113-2118 (1965).
- [33] V. N. Mineev and A. G. Ivanov, Electromotive force produced by shock compression of a substance, *Sov. Phys. Usp.*, **19**, No. 5, 400-419 (1976).
- [34] R. A. Graham, Shock-induced electrical activity in polymeric solids. a mechanically induced bond scission model, *The Journal of Physical Chemistry*, **83**, No. 23, 3048-3056(1979).
- [35] Paul Harris, Henri Noël Presles, The shock induced electrical polarization of water, *J. Chem. Phys.* **77**, 5157 (1982);
- [36] Paul. Harris, Mechanism for the shock polarization of dielectrics, *J. Appl. Phys.* **36**, No. 3, 739-741(1965).
- [37] J. Y. Wong, R. K. Linde, R. M. White, Electrical signals in dynamically stressed ionic crystals: a dislocation model, *J. Appl. Phys.* **40**, No. 10, 4137-4145(1969).
- [38] F. E. Allison, Shock-induced polarization in plastics. I. theory, *J. Appl. Phys.* **36**, No. 7, 2111-2113(1965).
- [39] R. J. Eichelberger and G. E. Hauver, in: *Les ondes de detonation*, P., 1961, p. 364.
- [40] Ronald K. Linde, William J. Murri, Donald G. Doran, Shock-induced electrical

polarization of alkali halides, *J. Appl. Phys.* **37**, No. 7, 2527-2532(1966).

[41] Paul Harris, Electromechanical shock waves, Technical Report No. 4532, July(1973).

[42] S. A. Bordzilovskii, S. M. Karakhanov, V. V. Polyudov, Investigation of the EMF of certain metal pairs under shock compression, 480-483(1972). (Novosibirsk. Translated from *Fizika Goreniya i Vzryva*, Vol. **8**, No. 4, pp. 586-590, October-December, 1972. Original article submitted May 3, 1972.)

[43] J. Grosnier, J. Jacquesson, and Migault, see <sup>[74]</sup>, p. 627.

[44] A. Migault and J. Jacquesson, in: Symposium of High Dynamic Pressure, Paris, 1967, p. 431.

[45] A. Migault and J. Jacquesson, *C. R. Ac. Sic. B* **264**, 507 (1967).

[46] A. I. Goncharov, S. P. Solovie, Shock-induced polarization of materials, *Combustion, Explosion, and Shock Waves*, **40**, No. 6, 658-662(2004).

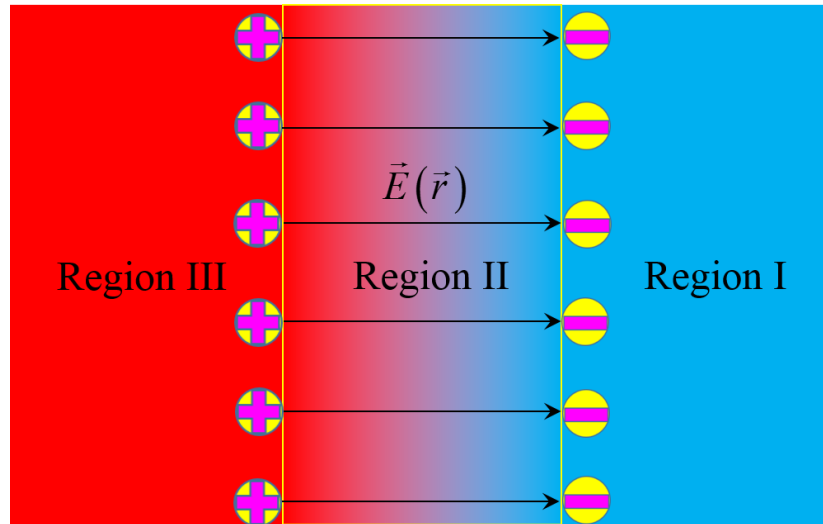


Figure 1. Schematic diagram of an electric field caused by *Yuheng Zhang effect* in materials: blue zones (Region I) and red zones (Region III) denote the unstrained regions and strained regions, respectively, and color gradient zone is strain gradient regions (Region II); stuffed circles with negative sign and stuffed circles with positive sign present electrons and ions; arrows in Region II give the possible electric field.



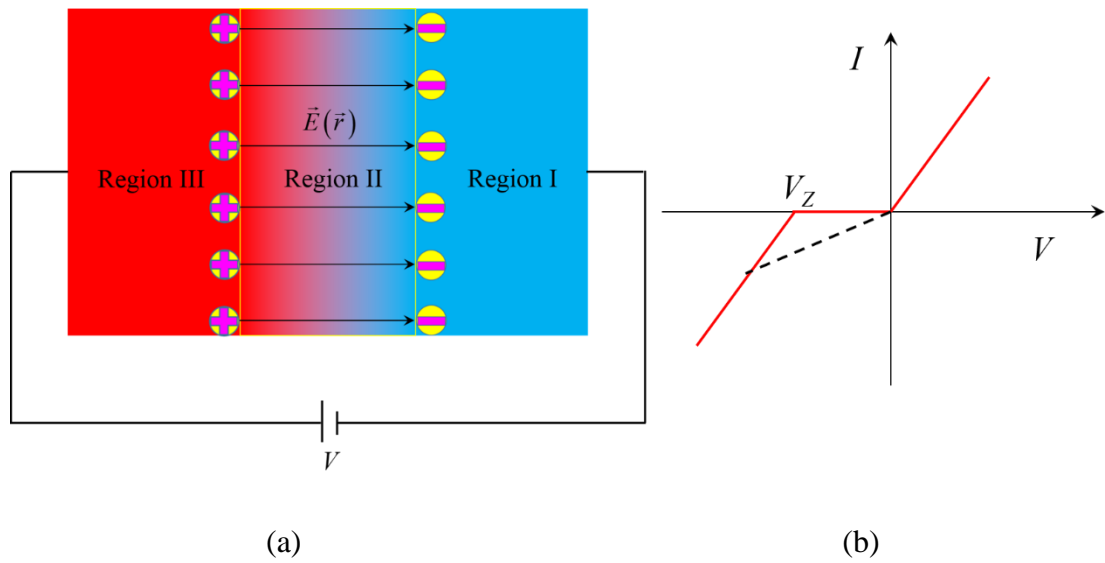
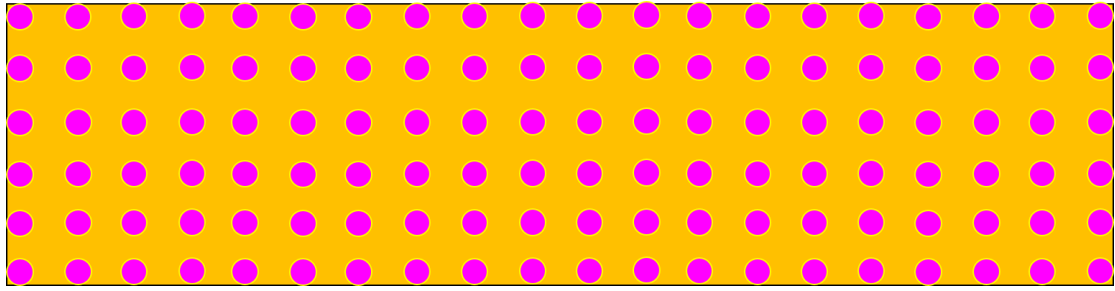
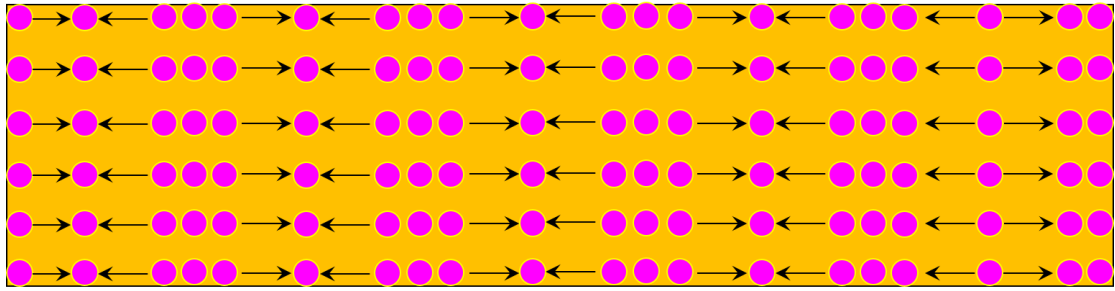


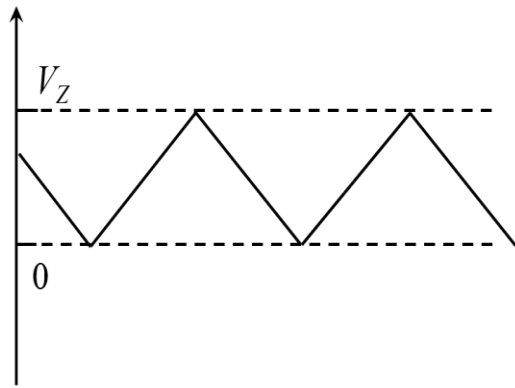
Figure 2. Sketch of metals and some narrow band semiconductors under strain gradient as well as their current-rectifying properties. (a) strained materials in electrical circuit with an applied external electrical voltage  $V$ ; Region I and Region III stand for the unstrained regions and strained regions, and Region II is strain gradient regions; stuffed circles with negative and positive signs present electrons and ions, respectively; arrows in Region II give the possible electric field distribution; (b) current-voltage ( $I$ - $V$ ) relation displayed by the metal and semiconductors under strain gradient,  $V_Z$  is *Yuheng Zhang potential* between Region I and Region III.



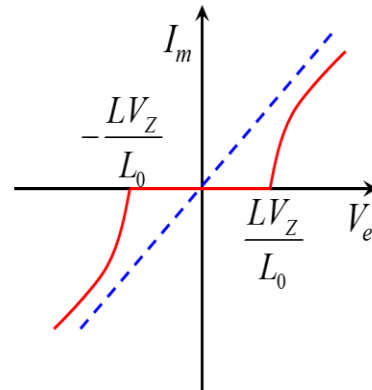
(a)



(b)



(c)



(d)

Figure 3. Schematic diagrams. (a) unstrained simple cubic (SC) crystalline lattice, the stuffed circles stand for ions; (b) periodically strained SC crystalline lattice, black arrows denotes the possible electric field distribution caused by *Yuheng Zhang effect*; (c) corresponding periodically electric potential for periodically strained crystalline lattice (b), where  $V_Z$  is the largest *Yuheng Zhang potential* in a period; (d) expected measured electrical current-voltage ( $I_m$ - $V_e$ ) relation and asymptotic line (blue dash line), where  $L$  is the crystal length,  $L_0$  is wavelength of the longitudinal wave.

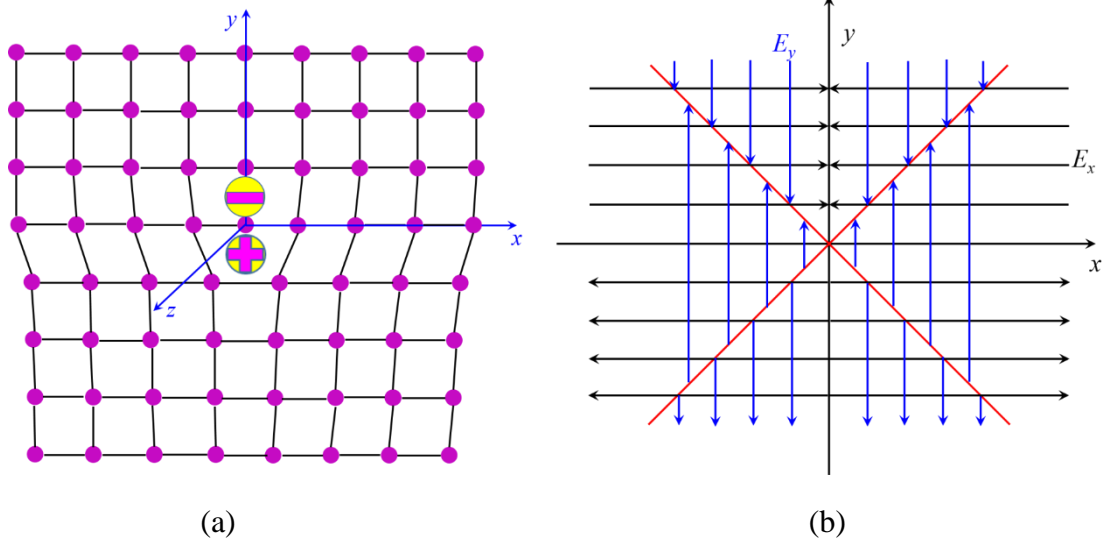
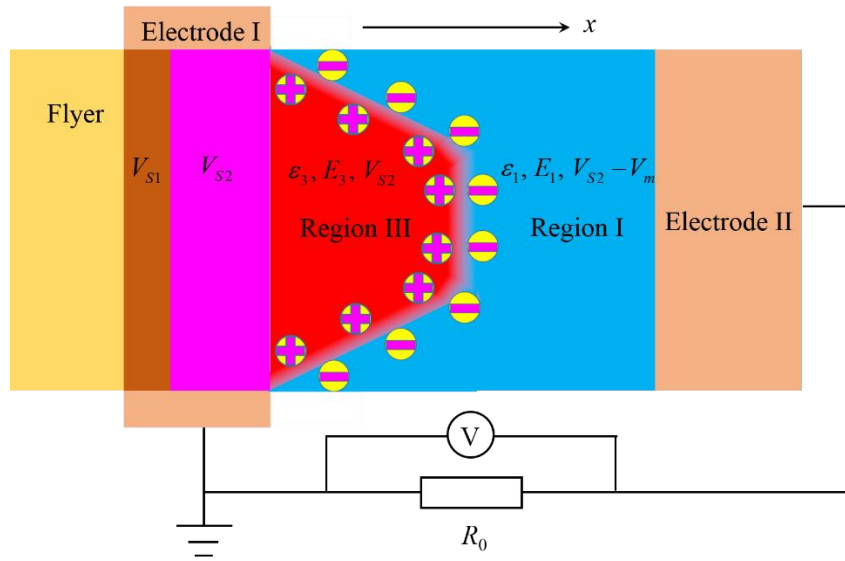
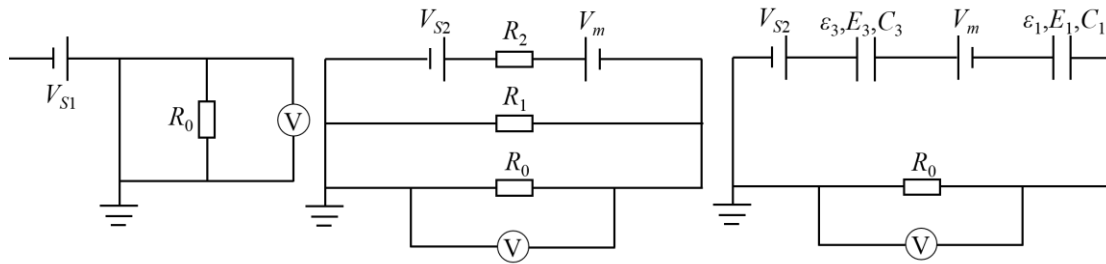


Figure 4. Sketch of an edge dislocation (a) and its electric field distribution (b) in an isotropic material. (a) stuffed purple circles denote atoms or ions, short black line are bonds, positive and negative stuffed yellow circles are line positive and negative charges, respectively; (b) black and blue arrows present direction of electric field  $E_x$  and  $E_y$ , but do not denote strength of these fields.



(a)



(b)

(c)

(d)

Figure 5. Schematic diagram of experimental setup (a) and corresponding equivalent electrical circuits (b), (c) and (d). In experimental setup (a), a flyer plate (left yellow zone) impacts an electrical screening plate which is also used as electrode I (left orange zones) and creates shock waves (left brown zone) which generate a *Yuheng Zhang potential*  $V_{S1}$  in electrode I, and (b) is the related electrical circuit. When arriving at interface between the screening plate and investigated material, the shock waves penetrate into the investigated material (light blue zones) and compress it (red zones). Across shock wave front (color gradient zone) in the material, another potential appears and is denoted as  $V_m$ . Electrode II, the same metal material as electrode I, is adopted and is connected to an external resistance  $R_0$ . Its voltage is the measured voltage. If the

investigated material is a metal, its equivalent electrical circuit is shown in (c), while electrical circuit for a dielectric material is shown in (d). The relative permittivity, electric field, capacitance, resistance for the dielectric material in shocked regions (red zones) and un-shocked regions (light blue zones) are  $\epsilon_3$ ,  $E_3$ ,  $C_3$  and  $\epsilon_1$ ,  $E_1$ ,  $C_1$ , respectively.

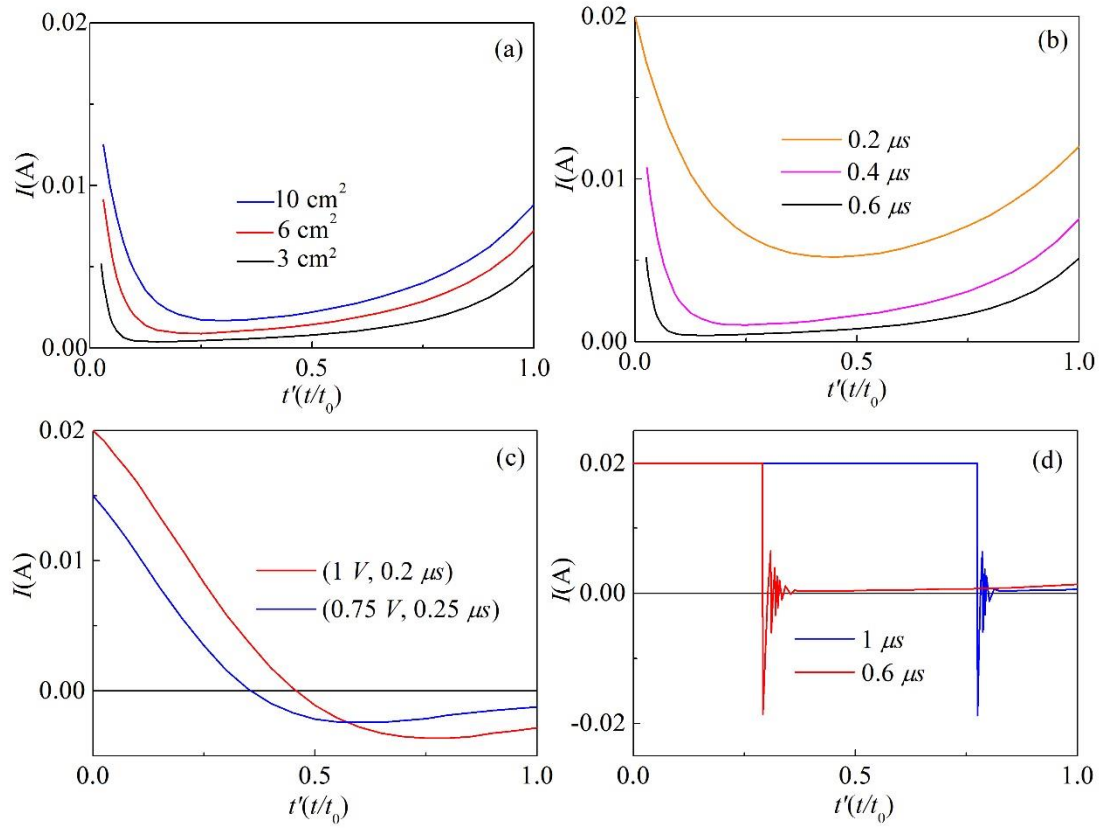


Figure 6. Calculated results for current-time relation ( $I$ - $t'$ ) for shock polarization. (a) sample area dependence of  $I$ - $t'$  relations under conditions  $\underline{V}=1$ ,  $\underline{R}=1$ ,  $\underline{t}=0.6$ ,  $k=6$ ,  $\underline{U}=3$  and  $\underline{\epsilon}=800$ , blue line ( $A=10 \text{ cm}^2$ ), red line ( $A=6 \text{ cm}^2$ ), black line ( $A=3 \text{ cm}^2$ ); (b) duration time dependence of  $I$ - $t'$  relations under conditions  $\underline{V}=1$ ,  $\underline{R}=1$ ,  $k=6$ ,  $\underline{U}=3$ ,  $\underline{A}=3$  and  $\underline{\epsilon}=800$ , orange line ( $t_0=0.2 \mu\text{s}$ ), magenta line ( $t_0=0.4 \mu\text{s}$ ), black line ( $t_0=0.6 \mu\text{s}$ ); (c) sign change of current under conditions  $\underline{R}=1$ ,  $k=0.2$ ,  $\underline{U}=3$ ,  $\underline{A}=3$  and  $\underline{\epsilon}=300$ , blue line ( $\underline{V}=0.75 \text{ V}$ ,  $t_0=0.25 \mu\text{s}$ ), red line ( $\underline{V}=1 \text{ V}$ ,  $t_0=0.2 \mu\text{s}$ ); (d) duration time dependence of a rectangular shape for  $I$ - $t'$  relations under conditions  $\underline{V}=1$ ,  $\underline{R}=1$ ,  $k=3$ ,  $\underline{U}=3$ ,  $\underline{A}=3$  and  $\underline{\epsilon}=300$ , blue line ( $t_0=1 \mu\text{s}$ ) and red line ( $t_0=0.6 \mu\text{s}$ ).

Table 1. For most materials with possible volume dependence of Fermi surface energy

$|\delta E_F(\vec{r})/\delta C_{ii}(\vec{r})|$  in the range 1~10 eV, the obtained flexoelectric coefficients are

listed as followings.

Relative permittivity $\varepsilon$	Flexoelectric coefficients $\mu_{ii} = \left  \frac{\varepsilon_0(\varepsilon-1)}{q} \frac{\delta E_F(\vec{r})}{\delta C_{ii}(\vec{r})} \right $
1-10	$10^{-11} \sim 10^{-9} \text{ C/m}$
10-100	$10^{-10} \sim 10^{-8} \text{ C/m}$
100-1000	$10^{-9} \sim 10^{-7} \text{ C/m}$
1000-10000	$10^{-8} \sim 10^{-6} \text{ C/m}$